High-Performance Ni-Co Sulfide Nanosheet-Nanotubes Grown on Ni Foam as a Binder Free Electrode for Supercapacitors

Jaffer Saddique, Xiaopeng Cheng, Huifeng Shi, Rui Wu and Yuefei Zhang *

Institute of Microstructure and Property of Advanced Materials, Beijing University of Technology, Beijing 100124, China
* Correspondence: yfzhang@bjut.edu.cn

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Abstract: The novel hierarchical Ni-Co sulfide nanosheet-nanotubes arrays were directly grown on Ni foam, as binder-free electrodes, have been successfully synthesized following a one-step facile hydrothermal method combined with a sulfide treatment. The initial value of the area capacitance achieved 2.28 F cm\(^{-2}\) at a current density of 1 mA cm\(^{-2}\). A high areal capacitance retention of 95.2% compared to activation-induced peak value is achieved after 3000 charge-discharge cycles, which is much better than counter Ni-Co oxide electrode (1.75 F cm\(^{-2}\) at 1 mA cm\(^{-2}\), 93.2% retention compared to activation induced peak value). The outstanding and excellent supercapacitive performance is ascribed to ion-exchange reaction, which induces a flexible hollow nanotube feature and show higher conductivity, compared with Ni-Co oxide NWs. Cyclic voltammetry (CV) and Electrochemical impedance spectra (EIS) results confirmed that the synthesized electrode contains the lowest resistance at high, and at lower frequency, leading to easy penetration of electrolytes and fast transportation of electrons inside the electrode. In this proposed work, a one-step hydrothermal method has been followed, and provided for the sulfide-induced, with a noticeable electrochemical performance of nickel cobaltite compounds and supplying a promising route for high-performance supercapacitor electrodes.

Keywords: Ni-Co sulfide; nanotubes; nickel foam; cation exchange; supercapacitors

1. Introduction

In recent years, sustainable energy and green energy systems get considerable attention to meet the increasing demand for energy throughout the world. Much effort has been invested to extend renewable energy sources [1,2], for example, rechargeable batteries and supercapacitors [3–5] are the main focus for research and get considerable attraction as energy-storage devices. Recently, supercapacitor received considerable attention, due to some of its interesting and beneficial advantages, like a long life time, high power density, and fast charge–discharge process. The electrochemical classification of a supercapacitor is based on charge-discharge progress and also on energy storage mechanisms, which can be classified into two types: Electrochemical double layer capacitors and pseudo-capacitors: (1) The electrochemical double layer capacitors can store energy directly by ion adsorption at the interface of the electrode/electrolyte and the main electrodes are carbon based material; (2) the pseudo-capacitors use the redox-active materials i.e., metal oxide-based material and conductive polymers, as electrode material allow the insertion redox reaction for both surface, as well at bulk state, and through fast surface redox reactions for storing energy [6,7].

The performance of supercapacitors is mainly connected with the properties of the electrode material. Generally, three types of electrode materials are being used in a supercapacitor: Carbonaceous
materials, transition metal oxides, and conducting polymer materials. Carbon-based materials, such as the typical materials of double-layer capacitors have a relatively lower specific capacitance and lower energy density, which hinders its practical application. In contrast, the metal oxides, as pseudo-capacitance materials, preserving much higher capacitance owing to the more charge storage at surface, get great interest and are widely investigated, due to their high theoretical capacitance for supercapacitors. The fabrication of these compounds, and especially oxides materials with diverse structures and morphologies, have attracted significant interest in material science and also at an industry level. Recently, efforts have also been made in the synthesis of binary component materials to upgrade the characteristics of materials and to enhance their characteristic properties beyond each individual component. Therefore, tremendous efforts have been devoted to the synthesis and fabrication of transition metal oxides, such as NiO [8–10], Fe₂O₃ [11,12], MnO₂ [13–15], Co₃O₄ [16–19], due to their high energy density, low cost, and environmental friendliness. Recently, cobalt sulfides and nickel sulfides, as new transitional metal sulfide electrode materials with high electrical conductivity and with superior electrochemical performances, have been investigated for pseudocapacitors [20,21]. However, the chemical and physical properties of transition metal sulfides are directly related to surface morphology, structure, and the size of the materials. The fabrication of these compounds, and especially sulfides material similar to oxides materials with diverse structures and morphologies, have attracted significant interest for developing electrode materials for energy storage devices [22]. Up until now, different types of structures, with a diverse morphology of Ni/Co sulfide, such as nanosheets [22], porous aerogels [23], urchin-like structures [24], have been reported. More importantly, the nanotube feature structures shows promising electrochemical performances with an increased rate of capability, as they provide an easy way for electrolyte penetration into active electrode material from the inner nanotube surface and in reducing the inactive electroactive surface area. As a result, a more electroactive surface area of electrode materials contribute to the electrochemical performances of the supercapacitors [25]. Additionally, in the comparison Ni-Co sulfide nanostructure, supported on conductive substrates, shows excellent pseudo-capacitive performances relatively to the nanostructure powders, because there is a chance that ion diffusion and electron transport are more quickly blocked during electrochemical performances when introducing the binders to make electrodes [26].

Based on the above considerations, in this paper, we adopted a one-step hydrothermal method, combined with thermal treatment, to synthesize Ni-Co sulfides nanotubes on Ni foam as a binder-free electrode for pseudocapacitors. The fabricated Ni-Co sulfides electrode on nickel foam possess a superior electrochemical performance (initial value is 2.28 F cm⁻²) to the pristine NiCo₂O₄ pattern (the initial value is 1.75 F cm⁻²) at current density of 1 mA cm⁻². Furthermore, a high areal capacitance 2.20 F cm⁻² with 95.2% retention, compared to the activation-induced peak value is achieved at 1 mA cm⁻² current density is achieved after 3000 charge-discharge cycles, which is much better than counter Ni-Co oxide electrode (a high areal capacitance 1.91 F cm⁻² with 93.2% retention compared to activation induced peak value). Herein, we adopted the hydrothermal method combined with thermal treatment for the synthesis of Ni-Co sulfide compounds to improve the capacitance. Our results indicate that the Ni-Co sulfide nanosheet-nanotubes arrays can act as high-performance electrode materials in supercapacitors.

2. Experimental Section

2.1. The Synthesis of NiCo₂O₄ Nanosheet-Nanowire Arrays

A one-step facile hydrothermal method was used for the fabrication of Cobalt Nickle Oxide nanoarrays on Ni foam combined with post thermal treatment. During the experiment the used reagents were of analytical grade and kept away from any further purification. To remove NiO layer, 2 M HCl was used carefully to clean the Ni foam, and then deionized water and absolute ethanol were used to wash the Ni foam -for 20 min respectively. Cobalt nitrate (Co(NO₃)₂·6H₂O), Nickelous nitrate (Co(NO₃)₂·6H₂O), ammonium fluoride (NH₄F), and urea were obtained from Beijing Chemical Reagent
Company. In a typical way, 0.38 g of Co(NO$_3$)$_2$·6H$_2$O, 0.19 g of Ni(NO$_3$)$_2$·6H$_2$O and 1.0 g of urea, 0.37 g of NH$_4$F were dissolved in 40 mL of deionized water and was subject to continuous magnetic stirring until a homogenous pink solution was achieved. Then the Ni foam (1 × 2 cm$^2$ in area) and the aqueous solution were transferred to 50 mL Teflon-lined stainless steel autoclave where the Ni foam of about 1 × 1 cm$^2$ in area was protected from solution contamination by uniformly coating the top with a polytetrafluoroethylene tape) and maintained at 120 °C for 6 h. Afterwards, a red thin product was obtained on Ni foam when cooled down to room temperature and subsequently washed with deionized water and ethanol for several time, dried at 80 °C for 6 h, and then finally calcined at 300 °C in air for 2 h.

2.2. The Synthesis of Ni–Co Sulfide Nanosheet-Nanotube Arrays

The Ni-Co sulfide nanosheet-nanotube arrays were obtained through a hydrothermal method by treating the as-obtained NiCo$_2$O$_4$ nanosheet-nanowire arrays with sodium sulphide (Na$_2$S). In detail, 1.20 g Na$_2$S·9H$_2$O was first dissolved in 40 mL deionized water and transferred into 50 mL Teflon container with the NiCo$_2$O$_4$ sample. Then the autoclave was sealed and kept in an oven at 120 °C. After 12 h, the autoclave was carried out and cooled down to the room temperature. Then, the product on Ni foam was obtained and subsequently washed with deionized water before being dried at 80 °C for 6 h.

2.3. Structural Characterization

The crystalline structure of the synthesized Ni-Co nano-arrays nanotube were examined by X-ray diffraction (XRD; Bruker D8) with Cu Ka radiation ($\lambda = 0.154$ nm). The state valence were characterized through X-ray photoelectron spectroscopy (XPS; Thermo Scientific Escalab 250Xi). The surface morphological studied were carried out by scanning electron microscopy (SEM; FEI Quanta 250) and transmission electron microscopy (TEM; JEOL, JEM-2010), elemental distribution were examined by energy-dispersive X-ray spectroscopy (EDS) system, while high resolution-transmission electron microscope (HRTEM; JEOL, JEM-2010 F) is for the investigation of lattice structure. The EDS mapping experiments were conducted by transmission electron microscopy (TEM, FEI Titan G2, operated at 300 kV). Areal capacitances were calculated using Equation (1) [27].

$$Ca = I \times \Delta t/(S \times \Delta V)$$

(1)

where constant discharge current were measured as I (mA), Δt (s) is the total discharge time, ΔV (V) is the potential drop during discharge and S is the electrochemical active area of the electrode.

2.4. Electrochemical Tests

A typical way is adopted for measuring the electrochemical performance and properties of the working electrode using three-electrode mode cells, prepared with a working electrode, while a platinum foil is used as the counter electrode, Hg/HgO is used as the reference electrode, and 6 M KOH aqueous solution is used as electrolyte. The electrochemical characteristics cyclic voltammetry (CV) were tested in the fixed potential window ranging between 0 and 0.6 V at different scan rates while that of electrochemical impedance spectroscopy (EIS) with the Ni-Co sulfides-Ni foam composite (1 cm$^2$ area) as the working electrode were measured at open circuit in the frequency range from 0.01 Hz to 100 kHz while applying potential 5 mV were carried out at CHI660D (Chenhua, Shanghai) work station., The Galvanostatic charge-discharge (GCD) of NiCo$_2$O$_4$ and the corresponding Ni-Co sulfide nickel foam were performed at 1 mV s$^{-1}$ and 1 mA cm$^{-2}$ and the cycle measurements both were performed on a LAND battery test system (LAND, Wuhan).
3. Results

The two-step environmental growth scheme of the Ni-Co sulfide nanosheet-nanotube on nickel foam is illustrated in Figure 1. Firstly, the NiCo$_2$O$_4$ precursor was firstly grown directly on Ni foam via following facile hydrothermal method, which is one of the most cost effective, environmentally friendly, and easy approachable non-toxic scientific methods for synthesizing controllable different morphology. Then, we carried out a calcination treatment so that the NiCo$_2$O$_4$ nanosheet-nanowire arrays on Ni foam can be converted hydrothermally to Ni-Co sulfide nanosheet-nanotube arrays counterparts by reacting with Na$_2$S, which is from attribution to the anion-exchange reaction between O$^{2-}$ and S$^{2-}$.

![Figure 1. Schematic illustration of the synthesis processes of Ni-Co sulfide nanosheet-nanotube arrays.](image)

Figure 2 presents the XRD pattern of the successful preparation of Ni-Co sulfide. The pure NiCo$_2$O$_4$ phase is observed in the pristine pattern, while after sulfurizing (anion-exchange) process, the final products can be indexed to mixture of Co$_4$S$_3$ (JCPDS No. 42-1448, $a = b = c = 9.387$) and Ni$_3$S$_2$ (JCPDS No. 44-1418, $a = b = 5.745$, $c = 7.135$), in which Co$_4$S$_3$ can be readily indexed to the (311), (400), (422), (511) and (440) and Ni$_3$S$_2$ can be indexed to the (101), (110), (021), (211), (213), (122), and (300). There are no other diffraction peaks, implying the full conversion of the precursor into Ni-Co sulfide during the hydrothermal process. XPS was performed to further confirm the prepared sample, while the well fitted spectra of Ni, Co, S (Figure S1) illustrates the successful anion-exchange reaction between O$^{2-}$ and S$^{2-}$. The Ni 2p and Co 2p spectra of Ni-Co sulfide (Figure S1a,b) are well fitted into Ni$^{2+}$, Ni$^{3+}$, Co$^{2+}$, and Co$^{3+}$ accompanied with shakeup satellite, respectively. While, S 2p spectra (Figure S1c) display strong peaks at 161.6 eV and 163 eV, which can be associated with S$^{2-}$ species in the composite and sulfur ions in low coordination at the surface [28].

![Figure 2. XRD pattern of as prepared Ni-Co sulfide arrays supported on Ni foam.](image)
To better elucidate the morphology, representative SEM images were carried out from the samples are given in Figure 3. The morphology of pristine Ni-Co oxide structures shown in Figure 3a,b demonstrates that nickel foam substrate surface was uniformly covered with a layer of uniform active material. The magnified SEM images (Figure 3c,d) show that large-scale, high dense and aligned Ni-Co sulfide arrays are supported on nickel foam substrate. EDS mapping of Ni-Co oxide and Ni-Co sulfide as shown in Figure S2. To demonstrate the morphology characters of Ni-Co sulfide arrays on Ni foam, the sulfide pattern was treated by focus ion beam irradiation. In the Figure 3e, the nanotubes in left part of the sample are wiped out by the Ga ion irradiation and the nanosheets feature is clearly exposed out of the arrays. And obviously the nanotubes (blue arrow) still left are connected with the nanosheet (yellow arrow), proving the formation of nanosheet-nanotubes feature of Ni-Co sulfide.

Figure 3. SEM images of (a,b) Ni-Co oxide, (c,d) Ni-Co sulfide, and (e) the Ni-Co sulfide partially treated by focus ion beam irradiation.
The closely structural information and morphology of the Ni-Co sulfide were investigated by TEM in Figure 4. A full view of a single hierarchical porous Ni-Co sulfide nanosheet-nanotube array in Figure 4a apparently shows the array consists of nanosheet below and nanotubes epitaxial growth from the nanosheet. The SAED pattern in the Figure 4b demonstrates the polycrystalline nature of the Ni-Co sulfide nanowire, the well-defined diffraction rings can be readily indexed to the \( \text{Co}_3\text{S}_4 \) \((220)\), \((222)\), \((511)\), \((440)\), \((800)\), and \( \text{Ni}_3\text{S}_2 \) \((021)\), \((202)\), \((211)\), \((122)\), respectively. Furthermore, the magnified TEM images in Figure 4c–e show that the Ni-Co sulfide nanowires are hollow internally with a sheath thickness of \(~15\, \text{nm}\) (Figure 4d). The nanotube and rough surface are very both effective in increasing the electroactive sites. The lattice fringes of HTTEM image in Figure 4f and g can be assigned to the \((220)\) and \((220)\) crystal plane of \( \text{Co}_3\text{S}_4 \), \((201)\), and \((202)\) plane of \( \text{Ni}_3\text{S}_2 \), respectively. The sample contains the elements of Co, Ni, and S evidenced by EDS pattern in Figure 4h, demonstrating that it is completely in sulfuration.

To demonstrate the electrochemical performance and properties of Ni-Co sulfide nanosheet-nanotube arrays, the comparative experiment was carried out between the NiCo\(_2\)O\(_4\) electrode and the Ni-Co sulfide electrode are shown and plotted in Figure 5. Firstly, a comparison of the cyclic voltammetry (CV) and galvanostatic charge-discharge (GCD) curves at \(1\, \text{mV}\, \text{s}^{-1}\) and \(1\, \text{mA}\, \text{cm}^{-2}\) of NiCo\(_2\)O\(_4\) and the corresponding Ni-Co sulfide nickel foam are shown in Figure 5a,b. Conspicuously, the integral area of the CV curve and the discharge time of Ni-Co sulfide is larger than Ni-Co oxides, implying excellent electrochemical energy storage capacity of the synthesized Ni-Co sulfide nanotube. Figure 5c shows the CV curves of Ni-Co sulfide at different scan rates from 1 to \(10\, \text{mV}\, \text{s}^{-1}\) ranging in a potential window between “0 to 0.6 V”. The pair redox peaks can clearly be observed and visible in all the CV curves leading to different pseudo-capacitive characteristics, during the electrochemical performance test and are directly related to the Faradaic redox reactions to M-O/M-O-OH [29], where M = Ni or Co. In Figure 5c, as the scan rate increases, the redox current increases; in the result, the corresponding anodic/cathodic peaks shifted towards higher and lower potential specifically at \(8\, \text{mV}\, \text{s}^{-1}\), and at \(10\, \text{mV}\, \text{s}^{-1}\), respectively with the large potential separation. This phenomenon is caused by the kinetic irreversibility in the redox process, due to Ohmic resistance and polarization behavior of the electrode [30]. The galvanostatic charge discharge curves of Ni-Co sulfide electrode with \(\text{Hg/HgO}\) reference electrode in the potential window ranging between 0 to 0.5 V at different selected current densities changing from 1 to \(8\, \text{mA}\, \text{cm}^{-2}\), as shown in Figure 5d. It can clearly observe that distinct plateau regions of the Ni-Co sulfide galvanostatic charge discharge are matching with the redox peaks in CV curves, indicates high pseudocapacitive characteristics of the Ni-Co sulfide. The cycle stability of supercapacitors for long-term is another critical issue to further evaluate the application of different structure electrodes. The cycling tests for the four synthesized electrodes were carried out by the repeated charging–discharging measurement at current density of \(1\, \text{mA}\, \text{cm}^{-2}\) over 3000 cycles. The long cycle performance and stability at a current density of \(1\, \text{mA}\, \text{cm}^{-2}\) are shown in Figure 5e, the Ni-Co sulfide nanotube achieved their peak \((2.31\, \text{F}\, \text{cm}^{-2})\) after 30 cycles and then became quite stable with 95.2% of the retained specific capacitance after 3000 cycles, while, the counter-part NiCo\(_2\)O\(_4\) nanowire array exhibited activation induced peak \((2.05\, \text{F}\, \text{cm}^{-2})\), which is attributed the enhanced conductivity and surface sites [31], then, followed by an obvious capacity decline to \(1.91\, \text{F}\, \text{cm}^{-2}\) after 3000 cycles. We can see that the cycling performance of Ni-Co oxide supercapacitor increases at the beginning and then decreases during the charging/discharging measurements. This is because the electrode cannot be completely wetted at the pristine cycles and has to go through a slow activation process for the electrode [32]. Furthermore, compared with previous work, as shown in Table S1, we clearly see that the Ni-Co sulfide nanosheet-nanotubes is a kind of excellent electrode material for supercapacitors.
The closely structural information and morphology of the Ni-Co sulfide were investigated by TEM in Figure 4. A full view of a single hierarchical porous Ni-Co sulfide nanosheet-nanotube array in Figure 4a apparently shows the array consists of nanosheet below and nanotubes epitaxial growth from the nanosheet. The SAED pattern in the Figure 4b demonstrates the polycrystalline nature of the Ni-Co sulfide nanowire, the well-defined diffraction rings can be readily indexed to the Co$_3$S$_4$ (220), (222), (400), (422), (511), (440), (800), and Ni$_3$S$_2$ (021), (202), (211), (122), respectively. Furthermore, the magnified TEM images in Figure 4c–e show that the Ni-Co sulfide nanowires are hollow internally with a sheath thickness of ~15 nm (Figure 4d). The nanotube and rough surface are very both effective in increasing the electroactive sites. The lattice fringes of HTTEM image in Figure 4f and g can be assigned to the (220) and (220) crystal plane of Co$_3$S$_4$, (201), and (202) plane of Ni$_3$S$_2$, respectively. The sample contains the elements of Co, Ni, and S evidenced by EDS pattern in Figure 4h, demonstrating that it is completely in sulfuration.

Furthermore, electrochemical impedance spectroscopy (EIS) test were measured to evaluate the electrochemical capacitance properties and stability of the Ni-Co sulfide electrode and is illustrated in Figure 5f, the tests were measured at open circuit in the frequency range from 0.01 Hz to 100 kHz while applying potential applying 5 mV. The obtained Nyquist plots of the Ni-Co oxide and sulfide electrodes contain semicircles part in high frequency and a linear part at low frequency region. The diameter of the semicircle and the straight short line of the obtained EIS spectra represent the total resistance during charge-transfer, the interfacial of the electrolyte resistance, and the resistance of the interface of the active material/current collector during electrochemical reaction. EIS result indicates the enhancement in the ion diffusion and effective electron transfer in the electrode charge-discharge process after sulfide, resulting in the excellent comprehensive electrochemical performance compared to Ni-Co oxide electrode.
Figure 5. Electrochemical performance of Ni-Co sulfide electrode and the pristine NiCo$_2$O$_4$ electrode. The electrochemical measurements were carried out in a 6 M KOH aqueous solution as electrolyte. (a) Cyclic voltammetry (CV) curves comparison, (b) Galvanostatic charge-discharge curves, (c) CV curve of Ni-Co sulfide electrode, (d) Galvanostatic charge-discharge measurements of Ni-Co sulfide electrode, (e) cycle performance comparison and (f) EIS curve comparison.

4. Conclusions

In summary, the novel growth of Ni-Co sulfide nanosheet-nanotube arrays as a binder free electrode on Ni foam have been successfully synthesized for supercapacitors, based on the anion-exchange reaction, by utilizing a one-step facile hydrothermal method combined with a subsequent low-cost sulfuration. The electrochemical performances of the transition metal oxides and sulfide electrodes materials are directly dependent on the shape, size, and morphological features obtained during the experimental process. The initial value of the area capacitance achieved 2.28 F cm$^{-2}$ of the present electrode at a current density of 1 mA cm$^{-2}$, a high areal capacitance retention of 95.2% is achieved compared to the activation induced peak value even after 3000 charge-discharge cycles, which is much better than counter Ni-Co oxide electrode (1.75 F cm$^{-2}$ at 1 mA cm$^{-2}$, 93.2% retention after 3000 charge-discharge cycles compared to activation induced peak value). The excellent super-capacitive performance is ascribed to the ion-exchange reaction, which induces the flexible hollow nanotube feature and higher conductivity, compared with Ni-Co oxide NWs.
These results evidently suggest the potential of the Ni-Co sulfide nanocomposite electrode for high-performance supercapacitors.

**Supplementary Materials:** The following are available online at [http://www.mdpi.com/2076-3417/9/15/3082/s1](http://www.mdpi.com/2076-3417/9/15/3082/s1), Figure S1: The XPS spectra of Ni-Co sulfide. (a) Ni 2p, (b) Co 2p, (c) S 2p, Figure S2: STEM-HAADF image and its corresponding EDS maps. (a-d) The sample of NiCoO$_2$, (e-h) the sample of Ni-Co sulfide. Table S1: Reference survey of related active material electrodes for electrochemical capacitors.

**Author Contributions:** J.S. and X.C. both contributed equally. Methodology and experiment, J.S., X.C. and R.W.; Writing-original draft preparation, X.C. and J.S.; writing-review and editing, H.S.; funding acquisition and supervision, Y.Z.

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**Conflicts of Interest:** The authors declare no conflict of interest.

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