- 1 Supplementary Material for: Dominant Effects of
- 2 Short-Chain Branching on the Initial Stage of
- 3 Nucleation and Formation of Tie Chains for
- 4 Bimodal Polyethylene as Revealed by Molecular
- 5 Dynamics Simulation
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- 14 1. Effect of the Box Size on the Crystallinity of Complex BPE Model
- 15 Fig. S1 shows the box size dependence of crystallinity (Xc) for HDPE model in Table 1 in the
- 16 manuscript. le* is the normalized distance from the end of the polymer chain to the closest edge
- 17 of the box. As observed, the value of Xc for HDPE model increased with the increase of the
- normalized distance l_e^* , up to about 6σ . Above 6σ , the value of Xc for HDPE model was almost
- stable at around 0.45. So in this paper, we chose the value of l_e^* as 25 σ to eliminate the effect of
- 20 box size on the crystallinity of all the models. Note that the side length of the box was much
- 21 larger than the normalized distance le*(when le* was set at about 25σ, the corresponding side
- length of the box was about 1100σ).

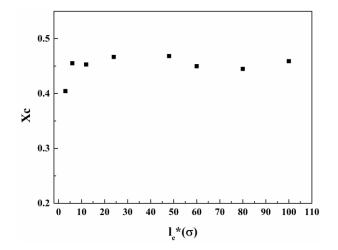


Figure S1. Box size dependence of crystallinity (Xc) for HDPE model in Table 1 in the manuscript. le* is the normalized distance from the end of the polymer chain to the closest edge of the box.

2. Comparison between Simulation Temperature and Real Experimental Temperature

In order to make a rough comparison with the real experimental melting temperature of BPE, we calculated the melting temperature of HDPE model (non-branched complex BPE model) in Table 1. The final semicrystalline state (state of the end of the simulation) of this model was then melted by heating it from $T^* = 3.0 - 8.0$ at a same heating rate $\Gamma^* = 2.5 \times 10^{-4} \tau^{-1}$, and the melting temperature of this model was $T^*_m = 7.5$. The corresponding experimental melting temperatures for BPE are in a range of 132 - 137 °C [1]. High simulation temperature $T^* = 8.0$ was corresponding to the real experimental temperature which was higher than the melting temperature of BPE.

3. References

1. Krishnaswamy, R.K.; Yang, Q.; Fernandez-Ballester, L.; Kornfield, J.A.

Effect of the distribution of short-chain branches on crystallization kinetics and mechanical properties of high-density polyethylene. *Macromolecules* **2008**, *41*, 1693-1704.

Other Figures and Tables

Table S1. Methylene sequence length (MSL) of the complex BPE model chains and the simple BPE

50 model chains

Branch Content (SCB/1000 backbone carbons)	Methylene Sequence Length (MSL)/CH2			
	Lb/10S	L/10Sb	Lb/1S	L/1Sb
1	476	334	1000	100
2	244	200	500	50
3	164	142	332	34
4	124	112	250	26
5	100	92	200	20
6	82	76	166	16
7	70	66	142	14
8	62	58	124	12

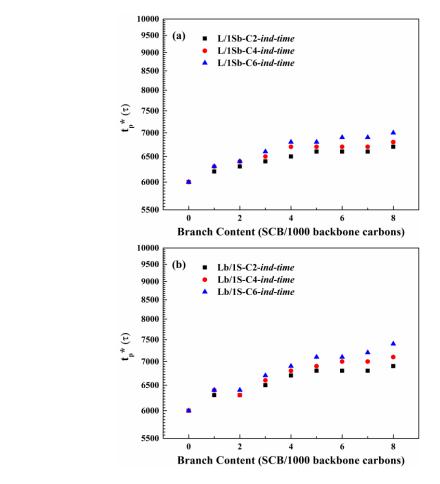
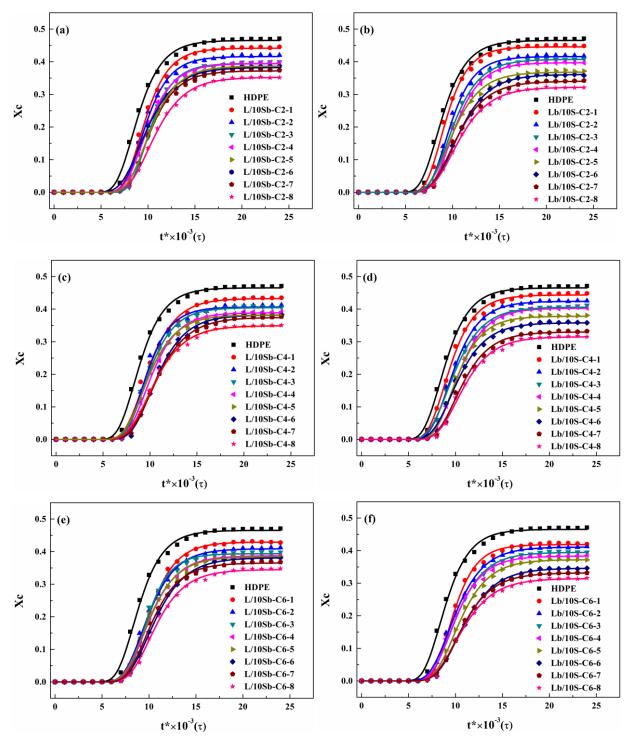


Figure S2. Branch length dependence of induction time (t_p^*) for (a) L/1Sb systems and (b) Lb/1S systems.



62 Figure S3. Crystallization curves of different branch content for (a) L/10Sb-C2, (b) Lb/10S-C2, (c) L/10Sb-

63 C4, (d) Lb/10S-C4, (e) L/10Sb-C6, (f) Lb/10S-C6 systems.

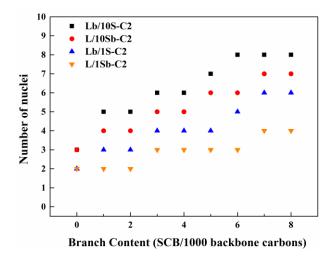


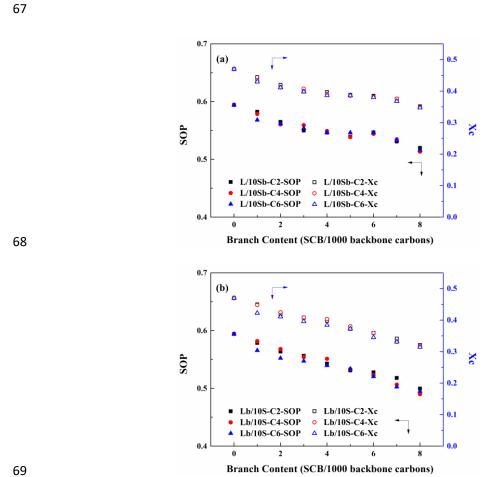
Figure S4. Number of nuclei of the ethyl branched complex BPE model chains and simple BPE model

chains at the end of the simulation.

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70 Figure S5. Branch length dependence of SOP and Xc for (a) L/10Sb systems and (b) Lb/10S systems.