

1. Introduction

The magnetic resonance imaging is currently one of the best diagnostic solutions adopted in clinic to identify different kinds of pathologies, due to the intrinsic high spatial resolution associated to the use of low energy radiation. More than 40% of the clinical practices require the administration to the patients of specific contrast agents. These probes are paramagnetic or super-paramagnetic compounds able to reduce the longitudinal ($T_1$) and/or the transversal relaxation time ($T_2$) of the water protons in tissues, thus increasing the sensitivity of the analysis, reducing possible artefacts, and improving the quality of the collected images [1–3]. Currently, the $T_1$-MRI probes used in clinics are mainly based on linear and cyclic Gd$^{3+}$-chelates, because they are characterized by good thermodynamic stability and kinetic inertness and often well tolerated by the patients [1]. Nevertheless, the amount of contrast agents required for the analysis is very high (from mM to M) and recently some studies reported on the accumulation of these probes in the cerebral membrane. To overcome these problems, three different strategies may be followed: (i) the design of novel paramagnetic probes based on less toxic metals (e.g., Mn$^{2+}$, Fe$^{3+}$, . . . ), (ii) the optimization of the Gd(III)-chelates to enhance the relaxivity performances; and (iii) the development of nanoparticles containing a large amount of paramagnetic sites, with high relaxivity at magnetic fields used in clinic (1.5–3 Tesla). This last strategy is particularly interesting because it favors a lowering of the detection limit of the MRI technique and a reduction of the contrast agent amount to administrate [4].

The paramagnetic nanoparticles proposed in the literature are composed using a diamagnetic inorganic support functionalized with Gd(III)-chelates, opportunely modified to promote their chemical attachment on the surface [5–10]; or they are completely inorganic [11,12]. In the last
case, the particles contain directly in the framework paramagnetic Gd\(^{3+}\) ions, often combined with other lanthanides with specific properties [11]. Recently, GdF\(_3\), NaGdF\(_4\), and Gd\(_2\)O\(_3\) systems were synthesized and proposed for diagnostic applications [11]. These nanoparticles typically require a careful design, aiming to reduce the particles size and to increase the surface hydrophilicity. For instance, Van Veggel et al., recently demonstrated that the best relaxometric performances could be achieved by decreasing the particles size to below 5 nm, with a consequent increase of the surface to volume ratio [13]. The selection of the capping agents to confine on the particles surface is another important topic. Particles with different anions and polymer groups were synthesized in order to increase the water suspendibility [14–16].

Gd\(_2\)O\(_3\) nanoparticles doped with luminescent lanthanides (Eu\(^{3+}\), Tb\(^{3+}\), Tm\(^{3+}\), and Er\(^{3+}\)) [17–21] have been extensively studied as potential optical imaging and MRI probes. Furthermore, because of the largest atomic number of Yb\(^{3+}\) among the lanthanides, mixed oxide nanoparticles containing Yb\(^{3+}\) and Gd\(^{3+}\) ions were proposed as a potential dual CT (computed tomography) and MRI probes, obtaining interesting results [22].

On the base of these considerations, we adopted in this work a low cost one-pot procedure for the preparation of Gd\(_2\)O\(_3\) nanoparticles, in order to reduce the reactions time and to eliminate the calcination steps that typically require high temperatures. Their surface was functionalized with citrate molecules able to stabilize the particles when dispersed in aqueous solution [23]. The idea to select citrate as chelating agent is motivated by the interesting results obtained for GdF\(_3\) nanoparticles [23]. It was demonstrated that citrate molecules confer both high hydrophilicity to the particles surface, thus improving the interaction of the metal ions exposed on the surface with the water molecules and high negative charge density, favoring the stability of the final aqueous suspensions over the time. In parallel, Gd\(_2\)O\(_3\) NPs were also doped with two different Yb\(^{3+}\) loading (5 and 10 mol %). The co-presence of Gd\(^{3+}\) and Yb\(^{3+}\) ions in the same particle can open the way to their possible use as dual MRI and CT contrast agents, as previously demonstrated for parent samples [22]. Specific attention was devoted to the investigation of their relaxometric properties as a function of the magnetic field and of their chemical stability in different conditions. These two aspects unfortunately are often missed in the literature and they require a deep comprehension to opportunely design very efficient MRI probes.

2. Results and Discussion

Gd\(_2\)O\(_3\) nanoparticles and the derivative materials containing different Yb\(^{3+}\) loading were prepared by adapting a synthetic precipitation procedure reported in the literature [24]. In detail, Gd\(^{3+}\) and Yb\(^{3+}\) precursors were dissolved in a few mL of triethylene glycol (TEG) that contained sodium hydroxide. TEG molecules work as chelating agent for the metal ions, thus limiting the particle growth [25]. The reaction was carried at 210 °C for 1.30 h in the presence of citric acid in low molar amount, in respect to Gd\(^{3+}\) salt (more details are reported in the experimental section). As clearly demonstrated in the literature, the citric acid plays two specific roles: (i) it limits the particles size by replacing the TEG molecules on the particles surface and (ii) it improves the hydrophilic character of the particles surface and the aqueous dispersion [23]. Nanoparticles with low (hereafter named Gd\(_2\)O\(_3\):Yb LL) and high Yb\(^{3+}\) loading (Gd\(_2\)O\(_3\):Yb HL) were also prepared by introducing a solution of 5 and 10 mol % of Yb\(^{3+}\) salt along with the Gd\(^{3+}\) precursor in the reaction. A schematic view of the Gd\(_2\)O\(_3\):Yb nanoparticles functionalized with citrate groups are reported in Figure 1A.
The structural properties of the samples were evaluated by an X-ray diffraction technique. The diffractogram of Gd2O3 NPs shows two wide bands centered at 29° and 44° 2θ (θ indicates the diffraction angle), attributed to the crystallographic planes 222 and 440, respectively (Figure 1B). This is typical of the body centered cubic structure, which has already been defined for these materials [26]. An additional component at low angles (ca. 22° 2θ) may be attributed to the organic fraction on the particles surface, which is in agreement with the literature data collected for parent samples. The X-ray profiles of the samples containing Yb3+ ions are completely comparable to that of Gd2O3, thus indicating that the incorporation in the structure of Yb3+ ions did not modify the structural features of these nanoparticles (Figure 1B). Furthermore, it is important to note that the presence of a wide band in the diffractogram of crystalline materials is typically associated to the nanometer nature of such samples. In light of these considerations, the average particles diameter (d) of all the samples was evaluated by applying the Debye–Scherrer equation (1) to the (222) reflection peak.

\[
d = \frac{k\lambda}{B_d \cos \theta}
\]

In the Debye–Scherrer equation, B_d is the full width at half the maximum intensity (FWHM) of the deconvoluted peak, \( \lambda \) is the X-ray wavelength, \( \theta \) represents the diffraction angle and \( k \) is the Scherrer’s constant that is 0.89 for spherical particles. The particle size estimated by X-ray diffraction (XRD) analysis for the Gd2O3 and derivative materials is approximately below 2.0 nm.

The chemical composition of the nanoparticles was estimated using elemental analysis (ICP-MS). The samples prior to the analysis were mineralized with concentrated nitric acid at a high temperature. Considering a density of 28.24 Gd/nm³ [27], the amount of Gd³⁺ ions into Gd2O3 was estimated to be around 400 per particle. This number decreases to 378 and 349 when increasing the Yb³⁺ loading in Gd2O3:Yb LL and Gd2O3:Yb HL, respectively. Further, 22 and 51 Yb³⁺ ions per particle that corresponded to 5.5 and 12.7 mol % were determined in the two samples at low and high Yb³⁺ loading. The amount of citrate molecules exposed on the nanoparticles surface was quantified by CHN analysis, resulting to 2.38 mmol/g for Gd2O3 sample and 2.25 and 2.08 mmol/g for Gd2O3:Yb LL and Gd2O3:Yb HL, respectively.

The presence of citrate on the nanoparticles surface was also confirmed through infrared spectroscopy (IR). IR spectra of Gd2O3 and the samples such as Yb³⁺ ions were collected at room temperature and appeared very similar (Figure 2).

Figure 1. (A) Schematic view of Gd2O3:Yb nanoparticles covered on the surface with citrate; (B) X-ray profiles of Gd2O3 (red), Gd2O3:Yb (LL) (green), and Gd2O3:Yb (HL) (blue).
The two main peaks at 1570 cm\(^{-1}\) and 1410 cm\(^{-1}\) are clearly detectable and assigned to the asymmetric and symmetric stretching modes of the deprotonated COO\(^-\) groups of citrate units. However, we must consider the band at 1570 cm\(^{-1}\) as a shoulder less intense at high wavenumbers, visible in all spectra that may be ascribed to the stretching vibrational mode of protonated COOH groups. The presence of this band suggests that a faction of pH-dependent protonated citric acid is also present on the particles surface.

The aqueous suspensions of the nanoparticles obtained directly by the synthesis procedure, without further modifications, were monitored by dynamic light scattering (DLS) analysis. The suspensions were visibly homogenous, as indicated in the digital photographs reported in Figure 3A. They appeared stable for ca, in which 1h without particles sedimentation did not require the use of surfactant or stabilizing agents to improve the particles suspension. Gd\(_2\)O\(_3\) suspension shows hydrodynamic diameter of ca. 40 nm because of a partial particles aggregation. The aggregation state was more pronounced for Gd\(_2\)O\(_3\):Yb LL and Gd\(_2\)O\(_3\):Yb HL suspension, with hydrodynamic radius of ca. 120 nm. A possible explanation of these differences in the aggregates size may be related to the different charge density exposed on the surface.

Figure 3. (A) Digital photograph of aqueous suspensions of Gd\(_2\)O\(_3\), Gd\(_2\)O\(_3\):Yb (LL) and Gd\(_2\)O\(_3\):Yb (HL) (from left to right); (B) dynamic light scattering (DLS) analysis of Gd\(_2\)O\(_3\) (red), Gd\(_2\)O\(_3\):Yb (LL) (green) and Gd\(_2\)O\(_3\):Yb (HL) (blue) suspensions.
To have more insights about this aspect, Z-potential analysis was performed on all samples in aqueous solutions, using the same experimental conditions as the DLS experiments. All nanoparticles showed a negative charge density with values of \(-14.0\) mV for Gd$_2$O$_3$ and \(-7.0\) and \(-10.7\) mV for Gd$_2$O$_3$:Yb LL and Gd$_2$O$_3$:Yb HL suspensions. These values are likely to be associated with the presence of the citrate carboxylate groups that are not involved in the coordination with Gd$^{3+}$ and/or Yb$^{3+}$ ions, which agrees with IR data. Differences in the amount of protonated and deprotonated citrate groups can be responsible for the different charge density on the surface of the samples and their aggregation behavior.

Finally, a preliminary relaxometric study was carried out in aqueous solution as a function of the magnetic field applied, in order to evaluate the mechanisms responsible of the magnetic interaction with water molecules. The longitudinal ($r_1$) and transverse ($r_2$) proton relaxivities defined the efficacy of 1 mM concentration of paramagnetic center to enhance the relaxation rates of the water protons. The relaxivity values for all the samples were measured at 40 MHz and 60 MHz at 310 K and neutral pH, along with the derivative $r_2/r_1$ ratio (Table 1). The Gd$^{3+}$ concentration in aqueous solutions was estimated by ICP-MS analysis.

<table>
<thead>
<tr>
<th>Samples</th>
<th>$r_1^{40}$ (mM$^{-1}$ s$^{-1}$)</th>
<th>$r_1^{60}$ (mM$^{-1}$ s$^{-1}$)</th>
<th>$r_2/r_1$ (40 MHz)</th>
<th>$r_2/r_1$ (60 MHz)</th>
</tr>
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<tbody>
<tr>
<td>Gd$_2$O$_3$</td>
<td>14.7</td>
<td>14.2</td>
<td>1.35</td>
<td>1.60</td>
</tr>
<tr>
<td>Gd$_2$O$_3$:Yb (LL)</td>
<td>14.7</td>
<td>14.7</td>
<td>1.39</td>
<td>1.92</td>
</tr>
<tr>
<td>Gd$_2$O$_3$:Yb (HL)</td>
<td>13.1</td>
<td>13.2</td>
<td>1.43</td>
<td>1.64</td>
</tr>
</tbody>
</table>

For all samples, the $r_2/r_1$ values do not change dramatically in the 20–60 MHz frequency range; they result to be below 2. This behavior suggests that these nanoparticles can be considered as positive MRI contrast agents [28,29]. The longitudinal relaxivity values calculated at high magnetic fields are in the 13–14 mM$^{-1}$ s$^{-1}$ range at 40 and 60 MHz, which is also the case for particles containing different Yb$^{3+}$ loading. Moreover, these values are in line with those calculated for parent Gd$_2$O$_3$ samples with comparable size [30] and much higher than those observed for commercial Gd$^{3+}$-chelates contrast agents [28]. The enhancement of the relaxivity value is governed by the dipolar coupling occurring between the paramagnetic ions and the protons of water molecules. This process involves two mechanisms: a short-range interaction with the water molecules directly coordinated to the metal ions (inner sphere contribution (IS)) or involved in H-bond with polar groups in close proximity to the paramagnetic ions (second sphere mechanism (SS)) and a long-range interaction with the bulk water molecules in proximity to the particles surface (outer sphere process (OS)) [31]. The contribution of each mechanism interaction is related to the position of the paramagnetic ions in the final nanoparticle. Metal ions distributed inside the particles core contributes to the relaxivity enhancement trough OS mechanism, whereas the paramagnetic centers confined in the surface are accessible to the water molecules and then they contribute to SS and/or IS mechanisms. It is known that when the inner sphere contribution is active, its effect is markedly high and it determines the final relaxivity value. Considering this in terms of Gd$_2$O$_3$ nanoparticles, the Gd$^{3+}$ ions are statistically distributed both inside and on the surface of the nanoparticles where the calculated relaxivity value is an average of conditions in which different mechanisms (IS, SS, and OS) are involved, which was additionally observed for GdF$_3$ and its parent materials [32].

To analyze in detail the role of the different mechanisms of interaction between the paramagnetic particles and the protons of water molecules, $^1$H relaxivity value of the samples dispensed in aqueous solution are measured as a function of the proton Larmor frequency (Nuclear Magnetic Resonance Dispersion (NMRD)) [33] at 310 K from 0.01 to 500 MHz (Figure 4).
The NMRD profiles are typical of macromolecular systems with slow mobility, with a plateau at low filed, followed by a dispersion from 1 to 10 MHz, and finally a hump at high magnetic fields with a maximum close to 60 MHz (1.5 T), as previously observed for Gd$_2$O$_3$ and GdF$_3$ nanoparticles with comparable size (Figure 4) [30]. It is important to note that the relaxivity values calculated at different magnetic fields are an average result considering that all the paramagnetic ions in the particle contribute in the same way to the relaxivity value. The best fit for the NMRD profile was obtained by applying the Solomon–Bloembergen–Morgan (SBM) [34] and Freed’s [35] equations for the inner- and outer-sphere proton relaxation mechanisms and considering also the second sphere contribution. Some parameters were maintained during the fitting: the distance of closest approach of the Gd$^{3+}$ ion and the inner sphere water molecule ($\tau_{rGdH}$) to 3.1 Å; and the distance between Gd$^{3+}$ and the inner sphere water protons ($\tau_{rGdH}$) to 4.0 Å; the distance between Gd$^{3+}$ ions exposed on the surface, was fixed to $3.1 \times 10^{-5}$ cm$^2$ s$^{-1}$. The more convincing results were obtained by considering one inner sphere water molecule ($q = 1$) coordinated to the Gd$^{3+}$ ions with $\tau_R$ of 0.35 ns for all samples. The differences in the aggregates size for the samples are limited and we did not observe a clear and direct relationship between the particles size and the $\tau_1$ values during the best-fitting procedure. Furthermore, the good overlapping of the $^1$H-NMRD profiles of Gd$_2$O$_3$ and Gd$_2$O$_3$:Yb LL testified these considerations. Nevertheless, the SS mechanism was present and it corresponded to the presence of one water molecule for Gd$_2$O$_3$ and Gd$_2$O$_3$:Yb (LL) samples and 0.75 waters for Gd$_2$O$_3$:Yb (HL), with a distance of 3.5 Å and $\tau_R'$ of 0.20 ns for the first two systems and 0.17 ns for the particles with high Yb$^{3+}$ loading. The electronic contributions correlated to the square of the zero-field splitting tensor, $\Delta^2$. The correlation time describing the modulation of the zero-field splitting ($\tau_V$) are comparable for all the samples and the relative values are reported in the Table 2.

Table 2. Selected best-fit parameters obtained from the analysis of the $1/T_1$ NMRD profiles (310 K) of all the nanoparticles.

<table>
<thead>
<tr>
<th>Samples</th>
<th>$\Delta^2$ (10$^{19}$ s$^{-2}$)</th>
<th>$\tau_V$ (ps)</th>
<th>$q$</th>
<th>$\tau_R$ (ns)</th>
<th>$q'$</th>
<th>$\tau_R'$ (ns)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Gd$_2$O$_3$</td>
<td>2.83 ± 0.17</td>
<td>37.0 ± 2.3</td>
<td>1</td>
<td>0.35 + 0.01</td>
<td>1</td>
<td>0.20 + 0.01</td>
</tr>
<tr>
<td>Gd$_2$O$_3$:Yb (LL)</td>
<td>2.77 ± 0.13</td>
<td>32.6 ± 1.6</td>
<td>1</td>
<td>0.35 + 0.01</td>
<td>1</td>
<td>0.20 + 0.02</td>
</tr>
<tr>
<td>Gd$_2$O$_3$:Yb (HL)</td>
<td>2.78 ± 0.21</td>
<td>32.8 ± 2.5</td>
<td>1</td>
<td>0.35 + 0.01</td>
<td>0.75</td>
<td>0.17 + 0.02</td>
</tr>
</tbody>
</table>

Figure 4. $1/T_1$ Nuclear Magnetic Resonance Dispersion (NMRD) profiles for Gd$_2$O$_3$ (red), Gd$_2$O$_3$:Yb (LL) (green), and Gd$_2$O$_3$:Yb (HL) (blue) at 310 K. The best-fit curves (solid lines) are calculated using the parameters of Table 2.
Finally, the stability of Gd₂O₃ nanoparticles was also monitored in reconstructed human serum (Seronorm) by measuring the longitudinal relaxation rate (R₁) values at 310 K and 40 MHz over the time. Despite a limited decrease of the relaxation rate after few minutes of ca, 10%, mainly ascribed to a possible alteration of the surface properties of the nanoparticles (i.e., replacement of the citrate ions by other anions present in the seronorm matrix), the values remained constant for 24 h. This test suggested that no detectable leaching of paramagnetic ions occurred in the matrix (Figure 5A).

In a second more severe test, the particles were treated with increased amount of a chelating agent, ethylenediammine tetracetic acid (EDTA). The behavior was different for the three samples. For Gd₂O₃, we assisted to a progressive decrease of R₁ value determined at 298 K and 40 MHz by increasing the EDTA amount. The EDTA/Gd³⁺ molar ratio of 0.5 was enough to promote the complete erosion of the nanoparticles with formation of the Gd(III)–EDTA chelate (Figure 5B). The particles bearing in the structure Yb³⁺ ions showed a completely different behavior. In the presence of limited amount of EDTA (EDTA/Gd³⁺ ratio of 0.3–0.5), the R₁ value increases and this was mainly evident for the sample with high Yb³⁺ loading. A further increase of the EDTA/Gd³⁺ molar ratio promoted a partial degradation of the nanoparticles with consequent decrease of the relaxivity values (Figure 5B). The partial increase of the relaxivity at the beginning was tentatively attributed to the complexation of the Yb³⁺ ions (not directly responsible of the relaxivity values observed for these NPs) on the surface and to the relative exposition of other internal Gd³⁺ ions, becoming more accessible to the water molecules.

Figure 5. (A) Stability over the time of Gd₂O₃ (red), Gd₂O₃:Yb (LL) (green) and Gd₂O₃:Yb (HL) (blue) in Seronorm suspension at 40 MHz and 310 K; (B) R₁ values (40 MHz, 298 K) for Gd₂O₃ (red), Gd₂O₃:Yb (LL) (green) and Gd₂O₃:Yb (blue) as a function of [ethylendiammine tetracetic acid (EDTA)]/[Gd³⁺] molar ratio.

3. Materials and Methods

**Gd₂O₃ NPs:** 2 mmol of Gd(NO₃)₃·6H₂O were dissolved in 10 mL of triethylene glycol (TEG). In parallel, 6 mmol of NaOH were added to other 10 mL of TEG. A third solution was prepared by dissolving 1.6 mmol of citric acid in 5 mL of TEG. All the solutions were stirred for 1 h at room temperature. In the second step, the first two solutions containing Gd(NO₃)₃ and NaOH were jointed together and maintained at 210 °C for 30 min. Then, the citric solution previously prepared was added and the final matrix was stirred at 210 °C for 1 h. A brown suspension was obtained. The solid phase containing Gd₂O₃ NPs was separated by centrifugation, and it was washed two times with 30 mL of ultrapure water. The particles were maintained in aqueous suspension in order to prevent particles aggregation.

**Gd₂O₃:Yb NPs:** A same procedure previously described for Gd₂O₃ was adopted for the preparation of mixed oxides containing low and high Ytterbium loading. The only difference is related to the molar amount of Gd³⁺ and Yb³⁺ precursor used in the reaction batch. In detail, for particles with lower Yb³⁺ amount (Gd₂O₃:Yb LL), 1.9 mmol of Gd(NO₃)₃·6H₂O, and 0.1 mmol
of \text{Yb(NO}_3\text{)}_3\cdot5\text{H}_2\text{O} \) were used. On the contrary, 1.8 mmol and 0.2 mmol of \text{Gd(NO}_3\text{)}_3\cdot6\text{H}_2\text{O} and \text{Yb(NO}_3\text{)}_3\cdot5\text{H}_2\text{O}, respectively, were dissolved in TEG solvent for the synthesis of the particles with higher \text{Yb}^{3+} \) loading (\text{Gd}_2\text{O}_3:\text{Yb HL}).

**Characterisation Methods**

The amount of \text{Gd}^{3+} \) and \text{Yb}^{3+} \) in the final materials was estimated using ICP-MS Thermo Scientific X5 Series (Waltham, MA, USA). The samples were mineralized via dissolution in \text{HNO}_3 \) (5 mL) at 120 °C for 24 h. The amount of citrate in the nanoparticles was quantified by using Euro EA CHNSO Analyzer of HEKAtech GmbH Company (Wegberg, Germany). XRD patterns were obtained on a ARL XTRA48 diffractometer (Portland, OR, USA) by using \text{Cu Kα} \) radiation (\lambda = 1.54062 Å). IR spectra were collected in air at 298 K in the range (4000–400 cm\(^{-1}\)) and with a resolution of 4 cm\(^{-1}\) by using a Bruker Equinox 55 spectrometer (Milano, Italy). The samples were mixed with \text{KBr} \) (10 wt %). DLS and Z-potential experiments were performed on a suspension of the particles in aqueous medium by using a Zetasizer NanoZS instrument (Malvern, UK) operating in the particle size range from 0.6 nm to 6 nm and equipped with a \text{He–Ne} \) laser (\lambda = 633 nm).

The water proton longitudinal relaxation rates were measured with a Stelar Spinmaster spectrometer (Pavia, Italy) operating from 20 to 70 MHz at 310 K. The standard inversion–recovery method was employed (16 experiments, 2 scans) with a typical 90° pulse width of 3.5 μs. The temperature was controlled with a Stelar VTC-91 airflow heater equipped with a copper–constantan thermocouple. The proton 1/\text{T}_1 \) NMRD profiles were measured on a fast field-cycling Stelar SmarTracer relaxometer over a continuum of magnetic field strengths from 0.00024 to 0.25 T (corresponding to 0.01–10 MHz proton Larmor frequencies). Additional data points in the range 20–70 MHz and at 500 MHz were obtained using a conventional spectrometer using WP 80 magnet interfaced with a STELAR console and Bruker NMR spectrometers, respectively.

4. **Conclusions**

In conclusion, \text{Gd}_2\text{O}_3 \) nanoparticles doped in the framework with different \text{Yb}^{3+} \) loading were successfully prepared through a fast and easily reproducible synthetic approach and they were decorated with hydrophilic citrate molecules. Nanoparticles that did not incorporate \text{Yb}^{3+} \) were also prepared as a reference material. In all cases, we obtained appreciable aqueous suspensions, which were stable and homogeneous without sedimentation. The samples showed relaxivity values at high magnetic fields, which were improved in respect to the clinically approved Gd(III)-chelates and in agreement with the results observed for comparable \text{Gd}_2\text{O}_3 \) and \text{GdF}_3 \) nanoparticles reported in the literature. \textsuperscript{1}H NMR relaxometry data indicated that the relaxivity values were mainly determined by the inner sphere contribution, but also second sphere water molecules H-bonded to the polar groups of citrate contribute to the final relaxivity. The nanoparticles were chemically stable in physiological medium (human serum). Furthermore, \text{Gd}_2\text{O}_3 \) samples doped with \text{Yb}^{3+} \) ions and with particles size distribution in aqueous solution centered at ca. Moreover, 120 nm (comparable to the samples here discussed) were tested both in vitro and in vivo as dual contrast agents and a toxicological study demonstrated the good biocompatibility and safety of these nanoparticles [22]. These features along with the co-presence of \text{Gd}^{3+} \) and \text{Yb}^{3+} \) ions in the same inorganic structure make these nanoparticles suitable for dual MRI-CT diagnostic analyses.

**Author Contributions:** Conceptualization, Investigation, Writing-Original Draft Preparation, Supervision, Project Administration: F.C.; Methodology, Formal Analysis, Resources, Data Curation, Writing-Review & Editing, Visualization, Funding Acquisition: F.C. and G.G.

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**Conflicts of Interest:** The authors declare no conflict of interest.
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