Abstract: Polymer matrix composites (PMC) have a competitive and dominant role in a lot of industries, like aerospace and automobiles. Short basalt fiber (SBF) is used to strengthen acrylonitrile–butadiene–styrene (ABS) polymers as a composite. The composite material is fabricated using injection molding with a new technique to obtain a uniform distribution for the ABS matrix at an elevated temperature range from 140 °C to 240 °C. Four types of specimen were produced according to the mechanically mixed amounts of SBF, which were (5, 10, 15, 20) wt %. The produced material was tested for tension, hardness and impact to measure the enhancement of the mechanical properties of the ABS only and the ABS reinforced by SBF composite. Wear tests were carried out using a pin on disc at a velocity of 57.5 m/s at three normal loads of 5, 10 and 15 kN. Tensile strength increased with up to 5 wt % of SBF, then decreased with an increasing amount of SBF reinforcement, while surface hardness increased with increasing SBF. The impact strength was found to degrade with the whole increment of SBF. Wear resistance increased with the increasing SBF reinforcement amount at all applied normal loads.

Keywords: acrylonitrile–butadiene–styrene (ABS); short basalt fiber (SBF); wear resistance; hardness; impact energy

1. Introduction

Composite materials nowadays play an increasing role in a lot of applications, such as the aerospace and automobile industry, and finally in biomedical applications. Acrylonitrile–butadiene–styrene (ABS) polymer plays an important role in many engineering plastics and composite materials, and has a rubber polybutadiene structure [1]. Natural fibers, like basalt fiber, nowadays plays a competitive role and receives special consideration compared to conventional glass and carbon fiber, because of their low cost and close mechanical and chemical properties [2–5].

Jiang et al. [6] investigated the effect of strengthening ABS with calcium carbonate particles of micro and nano size. They concluded that the microcrystalline cellulose/acrylonitrile–butadiene–
styrene (MCC/ABS) composite had a higher relative elastic modulus and lower relative tensile and impact strength when compared with ABS alone, while in Sept et al. [7], it was proved that particle size and heat treatment have little effect on the mechanical, thermal and physical properties of ABS.

Abdel-Haleem et al. [8] strengthen ABS using basalt fiber (BF). They used an injection molding machine to manufacture the BF/ABS composite with a varying wt % of BF. They investigated the effect of the basalt fiber content on the tensile and impact strength, and on hardness. They concluded that both tensile strength and Rockwell hardness improved with the increased wt % of BF, while the impact strength decreased.

Lopresto et al. [9] compared the mechanical performance of basalt fiber and glass fiber, and found that basalt fiber produces a higher Young’s modulus, compressive strength, and impact strength. They reported that low-cost basalt fiber is more competitive in glass fiber applications.

Difallah et al. [10] investigated the mechanical and tribological properties of ABS filled with graphite powder. It was observed that increasing the graphite content in the ABS matrix led to a drastic decrease in tensile strength, while it produced a good enhancement of the tribological properties and friction coefficient; this was possibly due to the fact that graphite powder is a solid lubricants.

Colombo et al. [11] used basalt fiber to strengthen the polymer matrix. They needed to study the effect of basalt fiber on the mechanical properties. The results showed that high mechanical properties were achieved for both tensile and compressive behaviors, moreover, the failure modes were found to be more tight and concise.

Jeevanantham et al. [12] experimentally investigated the tensile and flexural strength of polyurethane reinforced by basalt fiber composite laminate. The composite laminate was fabricated using a match plate mold under 1500 kg/cm$^2$ for 4 h at room temperature. They studied the effect of varying weight fractions of basalt fiber. It resulted in good tensile and flexural strength up to a certain weight fraction.

Chen et al. [13] carried out quasi-static and dynamic tests on unidirectional basalt-fiber-reinforced polymer (BFRP) to study the tensile properties of such a material. They observed that the tensile properties of BFRP were strain rate sensitive, they also extracted a good empirical formulation for predicting tensile properties. Khosravi and Farsani [14] attempted to enhance the mechanical properties of a unidirectional basalt fiber/epoxy composite using a modified nanoclay. Scalici et al. [15] carried out work on a basalt-fiber-reinforced composite using different vacuum-assisted impregnation techniques.

Wu et al. [16] investigated the durability of a basalt-fiber-reinforced epoxy composite in a different corrosive medium. It was found that the basalt fiber showed relatively strong resistance to water and salt corrosion, reasonable resistance to acid corrosion, and bad resistance in alkaline liquids.

Many studies have been carried out on continuous and short basalt fiber as a reinforcement phase [17–31].

Basalt fiber is natural fiber that is produced by basalt rock, this makes their chemical, physical and mechanical properties sensitive to the type of raw material, technology, and method of obtaining the final product [9,32]. Therefore, reinforcement plastic with this type of material needs a lot of investigation and studies.

Sudeepan et al. [33] studied the effect of varying micro-size zinc oxide (Zno) contents in a matrix of ABS polymer on the mechanical and tribological properties. It was concluded that tensile and flexural modulus and hardness generally increase with increasing filler content, whereas tensile and flexural strength increased up to 15 wt %, before decreasing again. They also reported the enhancement of the wear and friction coefficient with the filler content.

Zhang et al. [34] fabricated a short-basalt-fiber-reinforced polyamide product and filled it with MOS$_2$ and graphite contents using a hot press molding technique. The tribological properties of the composite product were investigated with a model ring-on block test rig, whereas the wear mechanism was studied based on scanning electron microscope examination. It was concluded that the tribological properties were improved with the filler contents.
Ha et al. [35] experimentally investigated the corrosion and tribological properties of basalt-fiber-reinforced composites. The corrosion effect was based on time and H2SO4 concentration. It was summarized that there was a general increase in specimen weight after corrosion in H2SO4 due to basalt fiber precipitate. The friction coefficient and wear behavior were irregular, while afterwards corrosion friction coefficient was two or three times greater.

The present study had three major main goals, which can be summarized as follows:

1. Fabrication of a novel composite material using ABS reinforced for the first time with short basalt fiber (SBF).
2. Measure the mechanical characteristic properties and durability of the new composite.
3. Measure the wear behaviors and wear mechanism of the novel composite.

The authors in [8] carried out a similar study, but using long basalt fiber, and without studying the tribological properties.

The paper structured as follows: in the first section the materials are described, in the second section the manufacturing technique for the composite sample is summarized, in the third section the mechanical and wear tests are explained, and finally the results and discussion are followed by the conclusion, where the best results are considered.

2. Materials and Methods

The work was performed using ABS that was supplied by LG Chem, Ltd., Hwachi-Dong, Yeosu-City, Jeonaranam-Do, 550-280, Korea. It is of RX710 grade, has a melting flow index of 21 g/10 min and a density of 1.04 g/cm³ [36]. The reinforcement of SBF was of 3:50 μm length and less than 6.5 μm in diameter, and was supplied by ROCKAL Co. (Cairo, Egypt) [37]. Figure 1 shows the raw materials of the composite product; their specifications are listed in Table 1. The basalt fiber as an inorganic material reinforcement has no homogeneity in melt [38]. It has many forms: fibrous, short or it may yarn [39].

![Composite raw material](image)

*Figure 1. Composite raw material. (A) acrylonitrile–butadiene–styrene (ABS); (B) Short basalt fiber (SBF).*

<table>
<thead>
<tr>
<th>Raw Material</th>
<th>Density (g·cm⁻³)</th>
<th>Melt Flow Index (g/10 min)</th>
<th>Particle Size (μm)</th>
<th>Diameter (μm)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Acrylonitrile–butadiene–styrene (ABS)</td>
<td>1.04</td>
<td>21</td>
<td>-</td>
<td>-</td>
</tr>
<tr>
<td>Short basalt fiber (SBF)</td>
<td>-</td>
<td>-</td>
<td>3:50</td>
<td>Less than 6.5</td>
</tr>
</tbody>
</table>
Sample Preparation

The specimens were manufactured using reciprocating screw injection molding (HAITIAN PL 1200, Haitian International Holdings Limited, Ningbo, China). The working range of temperatures of the barrel along the injection molding machine was 140 °C, 170 °C, 220 °C and 240 °C, as listed in Table 2. The ABS was dried at 12 °C for nearly 5 h to avoid possible moisture degradation. The SBF was mechanically mixed with ABS by (5, 10, 15, and 20) wt %. The direction of the polymer plastic flow was controlled to be unidirectional for all produced specimens, this is to avoid the weld line, which can lead to weakness and cracks. The mold of the injection molding machine is explained in [8]. The standard specimens were all 4 mm thickness for the tension test, 5 mm for the impact test and 6 mm for the wear test.

Table 2. Conditions of the injection molding machine.

<table>
<thead>
<tr>
<th>Parameter</th>
<th>First Stage</th>
<th>Second Stage</th>
<th>Third Stage</th>
<th>Fourth Stage</th>
<th>Fifth Stage</th>
</tr>
</thead>
<tbody>
<tr>
<td>Injection</td>
<td>140</td>
<td>140</td>
<td>140</td>
<td>130</td>
<td>130</td>
</tr>
<tr>
<td>Closing</td>
<td>80</td>
<td>80</td>
<td>80</td>
<td>30</td>
<td>100</td>
</tr>
<tr>
<td>Holding</td>
<td>100</td>
<td>100</td>
<td>100</td>
<td>-</td>
<td>-</td>
</tr>
<tr>
<td>Opening</td>
<td>60</td>
<td>60</td>
<td>60</td>
<td>60</td>
<td>60</td>
</tr>
<tr>
<td>Charging</td>
<td>100</td>
<td>100</td>
<td>100</td>
<td>-</td>
<td>-</td>
</tr>
</tbody>
</table>

3. Mechanical and Wear Testing

All the produced specimens of SBF-reinforced ABS composite were tested in tension according to the ASTM D3039/[40] at room temperature and 50% relative humidity. The tensile test was carried out using the one-ton capacity universal testing machine at 5 mm/min crosshead speed. The dog bone tensile test specimen is shown in Figure 2 with dimensions. The Charpy impact test was performed on all specimens accord to ASTM D6110-17 [41] at room temperature using a 5 kg hammer (see standard sample Figure 2b). The Rockwell hardness test, which is based on the Rockwell C-method, was carried out using a Digital Rockwell Hardness Tester, HRS-150 (Beijing United Test Co., Ltd., Beijing, China), the test was performed at room temperature according to the ASTM D785-08 standard [42]. The wear test was carried out on all produced specimens according to the ASTM G132-96 standard [43] at room temperature and 57.5 m/s using a pin on disc device, the counter face of the abrasive disc was of surface roughness (p1000). The weight loss was measured for each specimen at various normal load (5, 10 and 15) kN before and after the test. A microstructure examination of the fractured SBF-reinforced ABS composite with the tension and damage surface in the wear test were investigated using JEOL JSM.5500LV (JEOL Ltd., Tokyo, Japan) scanning electron microscopy at an acceleration voltage of 25 kV after immersion in nitrogen gas medium. The fractured and damaged surface was coated with a very thin layered of gold. Modes of failure were observed optically using a high-resolution digital camera for tension test samples.

Figure 2. Standard specimen (a) Tension test; (b) Impact test. (Unit: mm).
4. Results and Discussion

4.1. Tension Test

Figures 3 and 4a show a stress and strain diagram of SBF-reinforced ABS; it is clear that the composite tensile strength increases with the increasing wt % of the SBF up to 5 wt %, then it decreases. This trend might be due to the large area size and very weak interface [1]. The ductility of the newly-produced material drastically reduced with the increasing amount of SBF, as is shown in the percentage length elongation curve in Figure 4b; this can be attributed to the small area of SBF, which causes weak interfaces with the ABS [6]. With the lower content of reinforcement, there was a larger interfacial area, which led to the specimen being able to withstand a larger load capacity. The Young’s modulus increased up to 5 wt % then decreased, this may be due to the low strain when the strain was evaluated (see Table 3). The 3D plot in Figure 5 relates the elongation and tensile strength of the wt % of SBF. The failure modes for all tension test specimens were net tension with some surface microcracks, which is observed near the fracture process zone (see Figure 6). Scanning electron microscope (SEM) examinations show a uniformly-distributed matrix of ABS and smooth surface failure without microcracks (see Figure 7a). However, the surface becomes rough with the increasing wt % of ABS (Figure 7b–e); this can be attributed to the breaking of the bonds between the SBF and ABS, as these bonds cause fiber bridging and pull the matrix to a location of joining. Fiber breaking (marked with rectangular lines) at the interface between the ABS was observed, while fiber pull-out (marked with circular lines) clearly appeared. For the 20 wt % SBF, the fracture surface shows a random distribution of SBF, which means good mixing with the matrix. Each test was repeated five times and the stem diameter variations (SDV) was calculated and is listed in Table 3.

![Stress–strain diagram for short basalt fiber (SBF)-reinforced acrylonitrile–butadiene–styrene (ABS).](image-url)
Figure 4. Variation with wt % (a) Tensile strength; (b) Percent elongation.

Figure 5. A 3D curve relating the tensile test results with wt % of SBF reinforcement.

Figure 6. Failure modes in the tensile test specimens.
The relation between the Rockwell hardness and impact strength with the variation of the wt % of SBF is shown in Figure 11. A rough surface could be observed. The interaction between the SBF and ABS matrix was poor, with an increasing fiber content, therefore, impact load were analyzed using SEM photos, which are shown in Figure 10. It is observed that the increase, with an increasing amount of reinforcement phase [45–47]. The fracture surfaces under the two reasons for impact strength degradation, one is the weakness of interfacial bonding between brittle SBF, which resists penetration through the material's composite surface. The impact strength enhancement in a composite material, one increases the interfacial bond between the polymer and reinforcement, and the other adds a flexible and elastic phase that increases the impact [44]. There are there are two mechanisms for impact strength enhancement in a composite material, one increases the interfacial bond between the polymer and reinforcement, and the other adds a flexible and elastic phase that increases the impact [44]. There are two reasons for impact strength degradation, one is the weakness of interfacial bonding between brittle and high stiffness SBF and ABS, and the other is the cavitation or inner distance between the SBF increase, with an increasing amount of reinforcement phase [45–47]. The fracture surfaces under the impact load were analyzed using SEM photos, which are shown in Figure 10. It is observed that the interaction between the SBF and ABS matrix was poor, with an increasing fiber contents, therefore, a rough surface could be observed. The relation between the Rockwell hardness and impact strength with the variation of the wt % of SBF is shown in Figure 11.

Figure 7. Scanning electron microscope (SEM) microstructure examination of fracture plane (a) ABS; (b) 5 wt % SBF; (c) 10 wt %; (d) 15 wt %; (e) 20 wt %.

Table 3. Tensile test results.

<table>
<thead>
<tr>
<th>Samples</th>
<th>Tensile Strength (MPa)</th>
<th>SDV</th>
<th>Elongation at Break (%)</th>
<th>SDV</th>
<th>Young's Modulus (MPa)</th>
</tr>
</thead>
<tbody>
<tr>
<td>ABS</td>
<td>40.32</td>
<td>0.36</td>
<td>35.7</td>
<td>0.36</td>
<td>1450</td>
</tr>
<tr>
<td>ABS + 5%SBF</td>
<td>72.69</td>
<td>0.63</td>
<td>14.94</td>
<td>0.62</td>
<td>1481</td>
</tr>
<tr>
<td>ABS + 10%SBF</td>
<td>64.82</td>
<td>0.82</td>
<td>12.53</td>
<td>0.81</td>
<td>1099.7</td>
</tr>
<tr>
<td>ABS + 15%SBF</td>
<td>64.29</td>
<td>0.58</td>
<td>14.94</td>
<td>0.57</td>
<td>1117.2</td>
</tr>
<tr>
<td>ABS + 20%SBF</td>
<td>39.07</td>
<td>1.16</td>
<td>6.79</td>
<td>1.062</td>
<td>871.3</td>
</tr>
</tbody>
</table>

4.2. Hardness and Impact Tests

Figure 8 shows results of the Rockwell hardness test (RH). It is clearly illustrated that with the increasing wt % of SBF in the ABS matrix, the (RH) increased. This is due to the increasing amount of brittle SBF, which resists penetration through the material’s composite surface. The impact strength represented by the released energy stored in the material was reduced with the increasing wt % of SBF reinforcement (see Figure 9), due to decreasing ductility. There are two mechanisms for impact strength degradation, one is the weakness of interfacial bonding between brittle and high stiffness SBF and ABS, and the other is the cavitation or inner distance between the SBF increase, with an increasing amount of reinforcement phase [45–47]. The fracture surfaces under the impact load were analyzed using SEM photos, which are shown in Figure 10. It is observed that the interaction between the SBF and ABS matrix was poor, with an increasing fiber contents, therefore, a rough surface could be observed. The relation between the Rockwell hardness and impact strength with the variation of the wt % of SBF is shown in Figure 11.
Figure 8. Vickers hardness variation with wt % of ABS.

Figure 9. Impact fracture toughness variation with wt % SBF.

Figure 10. SEM microstructure examination of fracture plane (a) 5 wt %; (b) 10 wt %; (c) 15 wt %; (d) 20 wt % content SBF.
4.3. Wear Test

The reinforcement phase plays a great role in the tribological behavior and wear of the polymer material, as reported in [48]. The ABS plays an important role in a lot of applications, such as bearing and slide material. Figure 12 shows the weight loss variation with the increasing wt % of SBF content reinforcement. The SBF is a ceramic material that can be an abrasive agent. This abrasive agent enhances and increases the wear resistance of the composite material. Figure 13 shows a change of weight loss with the variation of normal applied loads and the variation of the wt % of the SBF. It can be clearly observed that with the increasing normal load, the weight loss increased, while with the increasing wt % of SBF, the weight loss decreased, which means that the wear resistance was enhanced. A total of 20 wt % of SBF results in a reduction in weight loss of nearly 78%, 82% and 84% for 5 kN, 10 kN, and 15 kN normal loads, respectively. The SEM micrograph of the damaged surface due to wear action is depicted in Figure 14 for ABS and 20 wt % SBF. The shear action is shown directly over the ABS (see Figure 14a), while for 20 wt % SBF the shear action interacted directly with the brittle fibers; therefore, an increasing wt % of SBF reduces the weight loss or wear rate (see Figure 13b). It was also found that the SBF is randomly distributed in the ABS matrix. There is some surface damage and breakage in the SBF fiber; this may be due to weak interfacial bonding between the SBF and ABS matrix.

Figure 11. Relation between the impact fracture and hardness with wt % SBF variation.

Figure 12. Wear resistance variation with wt % of SBF.
Figure 13. Wear resistance variation with normal load at a variation of wt % of SBF.

Figure 14. Shear mode in wear test. (a) ABS; (b) SBF.

5. Conclusions

Inorganic fillers have been used widely to enhance the mechanical and tribological properties of polymer products at lower cost. When SBF is used as a reinforcement in an ABS matrix, it is an inorganic material found to have weak interfacial bonding with the ABS. Increasing the wt % of the SBF content in the ABS matrix enhanced both the surface hardness and wear strength, while the weakness of the interfacial bonding between the fiber and the ABS led to decreased impact strength. Moreover, the tensile strength as well as the percent elongation in general decreased with the increasing wt % of SBF. It can be clearly concluded that the SBF is randomly distributed in the ABS matrix. The SBF failure mechanism may be fiber pull-out, fiber breakage and an absence of matrix carking. To enhance the tensile strength of a composite with an ABS base matrix, the SBF content should not exceed 5 wt %.

Author Contributions: Conceptualization, Validation, Formal Analysis, writing and Methodology, M.Y.A. and H.I.F., Software and Writing-Review & Editing, A.M.M.A. and M.D.

Funding: This research received no external funding.

Acknowledgments: Faculty of engineering energy at university of Aswan due to their continuous help during preforming these works.

Conflicts of Interest: The authors declare no conflict of interest.

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